

# Effect of the $\text{Al}_2\text{O}_3$ and BaO Addition on the Thermal and Physical Properties of Ternary Glass System ( $\text{B}_2\text{O}_3$ -BaO- $\text{Al}_2\text{O}_3$ )

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**Abstract:** In borate glasses, the main structural units are the  $[\text{BO}_3]$  triangles and  $[\text{BO}_4]$  tetradral which form different superstructural units like; boroxol rings, metaborate rings and chains, pentaborate, diborate, triborate and pyroborate. In this work, the Barium aluminoborate glasses were prepared. Some of properties were investigated by measure like density and chemical durability and the other by calculs. The dilatometric curves were determined and they revealed that the temperature of transition ( $T_g$ ) and softening ( $T_s$ ) and the dilatation coefficient increase by addition of  $\text{Al}_2\text{O}_3$  and BaO content.

**Key words:** Barium aluminoborate glass, density, chemical durability, dilatometric curves.

## Nomenclature

$\alpha$ :	Thermal dilation coefficient, $10^{-6}, \text{K}^{-1}$
$n_d$ :	Index of refraction
$d$ :	Optical dispersion
$E$ :	Longitudinal modulus of elasticity, kbar
$\sigma$ :	Surface tension $10^{-3}, \text{N/m}$
$\rho$ :	Densities, $\text{g}\cdot\text{cm}^{-3}$
$\sigma$ :	Compressive stress ( $\sigma_c$ ) or with traction ( $\sigma_t$ ), $\text{MN/m}^2$
$C_p$ :	Heat capacity, $\text{J/gK}$
$\lambda_c$ :	Thermal conductivity, $\text{W/mK}$
$\epsilon$ :	Electric permittivity

## 1. Introduction

In special glass systems, the chemical composition plays an important role in determining properties of the glass. The components of glass are distributed into three categories: network formers, network modifiers and intermediate species, which falls somewhere between network formers and modifiers

and may substitute for a network former in the glassy state. The higher valence cations such as  $\text{Al}^{3+}$  are commonly used as intermediate species [1]. In borate glasses, the main structural units are the  $[\text{BO}_3]$  triangles and  $[\text{BO}_4]$  tetradral which form different superstructural units like; boroxol rings, metaborate rings and chains, pentaborate, diborate, triborate and pyroborate [2]. Also, in the pure  $\text{B}_2\text{O}_3$  glass consists of  $[\text{BO}_3]$  groups and with increasing alkali concentration, the first incorporation of alkali oxide leads to the coordination shift  $[\text{BO}_3]$  to  $[\text{BO}^{4-}]$  with alkali ions compensating the charge of the  $[\text{BO}^{4-}]$  tetrahedra. A strengthening of the structure occurs, since the points of polyhedra linkage rise from three to four. At higher alkali concentrations, the structure weakened again due to the formation of  $[\text{BO}_3]$  groups with nonbridging oxygens. This double change of coordination number explains minima or maxima of some properties which occur with increasing alkali concentration. Furthermore, this anomaly depends also on the temperature. According to Dietzel [3], the

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boron anomaly should not exist in glasses at temperatures above 1,000 °C, i.e., above this temperature only  $[\text{BO}_3]$  groups occur in the melt. The association of  $[\text{BO}_3]$  planar triangles with oxygen atoms into  $[\text{BO}_4^-]$  groups should, however, occur at lower temperatures. Aluminum oxide acts as an intermediate in glass. Intermediates have a mid-position between network-formers and network-modifiers. Aluminum may either form tetrahedra and so reinforce the network (coordination number 4) or loosen the network in analogy to network-modifiers (coordination number 6). The ionic radius of barium ions is much larger than of aluminum oxide and boron oxide, which results in the coordination number 8. By contrast, the field strength of barium is much smaller resulting in the behaviour as network-modifier. In the  $\text{BaO-Al}_2\text{O}_3\text{-B}_2\text{O}_3$  glass system, it was reported that  $\text{Al}_2\text{O}_3$  behaves as  $\text{AlO}_4$  and  $\text{AlO}_6$  units in the glass structure [4]. It assumes that aluminium ions enter the structure in the form of three tetrahedral  $\text{BO}_4$  and/or  $\text{Al}_2\text{O}_3$  in the form of  $\text{AlO}_4$  having an oxygen in common [5]. Owen [6] proposed for the glass systems of  $\text{MO-Al}_2\text{O}_3\text{-B}_2\text{O}_3$ ; ( $\text{M} = \text{Sr}, \text{Ca}$  and  $\text{Ba}$ ) that some of the  $\text{M}$  atom associate themselves with  $\text{Al}_2\text{O}_3$  forming  $\text{AlO}_4$  and the rest act with  $\text{B}_2\text{O}_3$  producing  $\text{BO}_4$  or non-bridging oxygen ions. The system  $\text{BaO-Al}_2\text{O}_3\text{-B}_2\text{O}_3$ , have also a negative thermal expansion coefficient, and therefore it is a potential candidate of a zero-expansion material [7, 8]. First research of this system with regard to the thermal expansion coefficient was carried out by MacDowell in 1989 [9, 10].

In the present study some properties of  $\text{BaO-Al}_2\text{O}_3\text{-B}_2\text{O}_3$  glass system were investigated by measure like density and chemical durability and the other by calculs. The dilatometric curves were determined and they revealed that the temperature of transition ( $T_g$ ) and softening ( $T_s$ ) and dilatation coefficient increases by addition of  $\text{Al}_2\text{O}_3$  and  $\text{BaO}$  content.

## 2. Experiments

### 2.1 Preparation of the Glass Samples

The glasses selected were prepared starting from the following chemical raw materials; barium carbonate, orthoboric acid and aluminum oxide. The finely crushed mixture was then placed in a platinum crucible and transferred to an electric furnace at temperature ranging from 1,300 °C to 1,400 °C with a stage for 1.5 h. The liquid was then cast in a graphite mold preheated to approximately 250 °C to limit the thermal shocks during hardening. The samples were annealing then at 350 °C for 1 h. The compositions of studied glasses are given in table 1.

### 2.2 DTA and TGA Analysis

The apparatus used was a simultaneous thermal analysis apparatus type STA 449C. Jupiter, it can give the differential variations in temperatures, changes in weight during treatment and thermal enthalpies exchanged. It works at high temperature furnace with protective tube of  $\text{Al}_2\text{O}_3$  and temperature range 25 to 1,550 °C. The type of the thermocouple used is Pt/Pt Rh.

The glass transition temperature ( $T_g$ ) was determined from the second endothermic peak of DTA curve whereas the crystallization temperatures ( $T_c$ ) was determined by the first exothermic peak of DTA curve [5].

### 2.3 Dilatometric Analysis

The expansion curves of samples were determined using a dilatometer DIL 402C (Materials Mineral Composite Laboratory (MMCL-Boumerdes-Algeria) at an average speed of heating of  $5 \text{ K}\cdot\text{min}^{-1}$ . The sample had a rectangular shape with an 8 mm width and a 20-25 mm length.

The glass transition temperature ( $T_g$ ) was determined from the expansion curve using the interception method, whereas the softening temperatures ( $T_s$ ) was determined by the maximum temperature of expansion curve [5, 7, 8].

**Table 1** Chemical composition of studied B<sub>2</sub>O<sub>3</sub>-BaO-Al<sub>2</sub>O<sub>3</sub> glasses system.

	Composition (weight %)			Composition (molaire %)		
	B <sub>2</sub> O <sub>3</sub>	BaO	Al <sub>2</sub> O <sub>3</sub>	B <sub>2</sub> O <sub>3</sub>	BaO	Al <sub>2</sub> O <sub>3</sub>
G <sub>1</sub>	95	5	-	97.70	2.29	-
G <sub>2</sub>	85	10	5	91.73	4.88	3.76
G <sub>3</sub>	75	15	10	84.92	7.69	7.77
G <sub>4</sub>	65	20	15	77.5	10.83	12.25

**Table 2** Values obtained of  $T_g$  and  $T_s$  of (BBA) glass system.

Samples	$T_g$ (°C)	$T_s$ (°C)
G <sub>1</sub>	312.3	335.5
G <sub>2</sub>	387.7	417.1
G <sub>3</sub>	422.8	443.8
G <sub>4</sub>	512.8	536.4

**Table 3** Values obtained of density ( $\rho$ ) and molecular volume ( $V_m$ ) of BBA glass system.

Samples	$\rho$ (g/cm <sup>3</sup> )	$V_m$ (cm <sup>3</sup> /mol)
G <sub>1</sub>	1.850	38.665
G <sub>2</sub>	2.143	35.081
G <sub>3</sub>	2.432	32.416
G <sub>4</sub>	2.561	32.391

## 2.4 Density Measurements

The densities were determined out using Archimedes' method with xylene as an immersion fluid. The relative error in these measurements was about  $\pm 0.03 \text{ g}\cdot\text{cm}^{-3}$  and the molar volume  $V_m$  was calculated from the molecular weight  $M$  and the density  $\rho$  according to the relation:  $V_m = M/\rho$ . The density results are illustrated in Table 3.

## 2.5 Determination of Chemical Durability

The chemical treatment of the samples under the various conditions (neutral, acidic and basic) was carried and the weight loss of samples was determined.

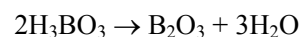
## 2.6 Theoretical Calculs of Properties

In the structure of glass, the various components contribute a share defined in the effect of certain properties. There would be thus a possibility of calculating by means of additive formulas these properties to leave the composition [11].

# 3. Results and Discussion

## 3.1 DTA and TGA Analysis

The (DTA) curves of these glasses, show endothermic peaks between 100 and 200 °C (Fig. 1), knowing that towards the temperature 100 °C, the boric acid starts to lose its water, giving initially metaboric acid, and by dehydration supplements with 300 °C, it forms vitreous boric anhydride according to the following relation:



This is accompanied by an important loss by mass presented in the relative curves TGA, however this loss of mass decreases with the reduction in the content of B<sub>2</sub>O<sub>3</sub> (of G<sub>2</sub> towards G<sub>4</sub>). The second endothermic peaks which this locate at various temperatures for various glasses represent the temperatures of vitreous transitions, their values increase with the increase in the oxide Al<sub>2</sub>O<sub>3</sub> and BaO.

In parallel, various first exothermic peaks corresponding to the beginning of crystallization of the vitreous samples which show also an increase in the values with the increase in the oxide Al<sub>2</sub>O<sub>3</sub> and BaO. Thus, the addition of the oxides Al<sub>2</sub>O<sub>3</sub> and BaO in the systems of glasses (BBA) increases the temperatures of vitreous transition and the temperatures from crystallization of these glasses.

## 3.2 Dilatometric Analysis

The obtained curves of thermal dilation are represented in the Fig. 2. For glass G<sub>1</sub> the thermal curve of dilation is representative of glass separated in two phases, because there are two glass transition temperatures represented by two points of inflection on the curve. The B<sub>2</sub>O<sub>3</sub>-BaO binary system is known by

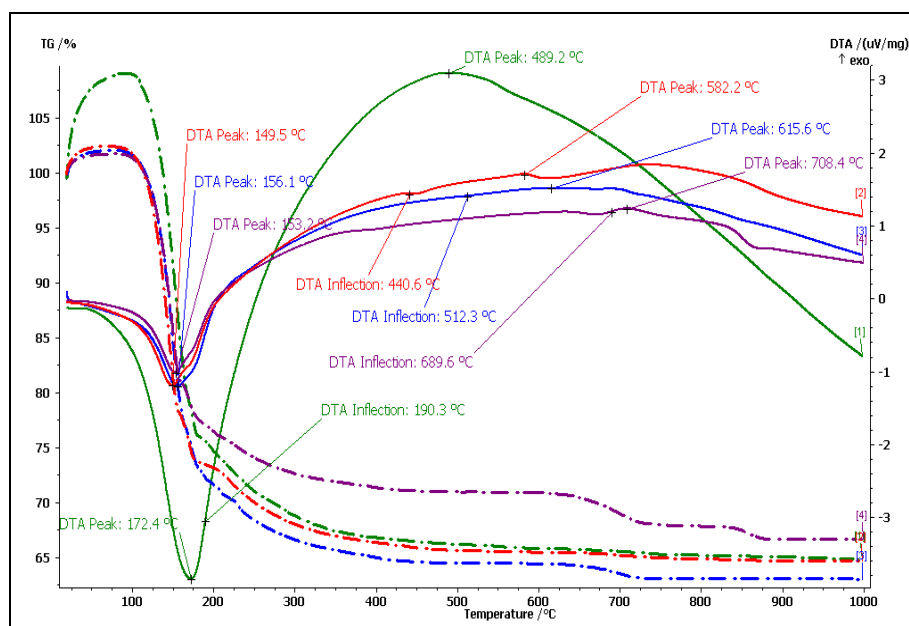


Fig. 1 DTA and TGA curves for obtained (BBA) glass system.

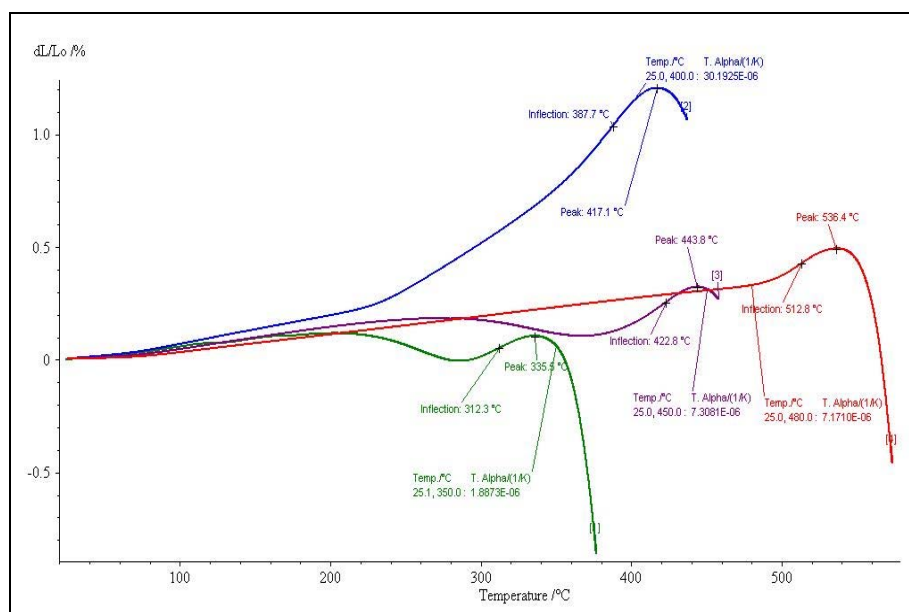


Fig. 2 Expansion curves of for obtained (BBA) glass system.

its mechanism of separation of phase [12, 13]. The G1 sample obtained has a translucent white aspect. With the addition of five percent of alumina (5%  $\text{Al}_2\text{O}_3$ ), the phase separation decreases and the other glasses are homogeneous.

The temperatures of vitreous transition  $T_g$  and softening temperatures  $T_s$  increase by G1 towards G4 with the addition of  $\text{Al}_2\text{O}_3$  and BaO, because these the last two oxides are known by their role to increase the

viscosity of the molten bath but thermal dilation increases by G1 towards G4. Generally, glasses with the lower expansion coefficient have higher transition and softening temperatures and vice versa [11]. However, it can be seen in Fig. 2 that the expansion increases and the  $T_g$  and  $T_s$  temperatures increase with additions of  $\text{Al}_2\text{O}_3$  and BaO oxides.

This is with the molar report/ratio  $\text{Al}_2\text{O}_3/\text{B}_2\text{O}_3$  which increases by G1 towards G4, or the  $\text{B}^{3+}$  ions go

to join with  $\text{Al}_2\text{O}_3$  by forming the tetrahedrons  $\text{AlO}_4$  and the remainder will act with the  $\text{B}_2\text{O}_3$  by producing not bridging oxygen (NBO) what has to weaken the vitreous network and dilation has to increase. Substitution of BaO for  $\text{B}_2\text{O}_3$  in the studied glass seems to lead to the conversion of NBO'S formation.

### 3.3 Density and Molar Volume

Density variation and molar volume for obtained (BBA) glass system as a function of  $\text{Al}_2\text{O}_3$  mol% are given in Fig. 3a. Adding the BaO and  $\text{Al}_2\text{O}_3$  oxides increase density: alkali earth ions (Ba) fill the free volume and  $\text{Al}_2\text{O}_3$  behaves as  $\text{AlO}_4$  and  $\text{AlO}_6$  units in the glass structure so the molar volume decreases (Fig. 3b).

### 3.4 Chemical Durability

Results illustrated in Table 4 showed that the samples exhibited an increase in durability under the three media conditions studied (neutral, acidic and basic) with increase of aluminium content. Chemical

attack decreases with increasing the aluminum content in all medium. The reactions between the ion  $\text{H}^+$  and the “acid” network can be neglected since the components are packed too strongly to allow for any possibility of migration in the network to take place. On the other hand, the network modifiers have a certain freedom of displacement through vacuum and also the ability to pass through the solution that surrounds them if this vacuum borders the solution. The attack by the alkaline solutions on the other hand is governed by another mechanism.

The  $\text{OH}^-$  ion is in the case the determining factor, because its ability to react with the network. As a consequence, it appears that this result in a network division, which under certain conditions can result a complete dissolution of glass. It also appears that the solubility of glass increases under basic pH conditions [14]. When comparing the ratio of attack by acid versus water, this attack decreases with increasing  $\text{Al}_2\text{O}_3$  content, which also reinforced the structure and reduces the chance of network division and destruction.

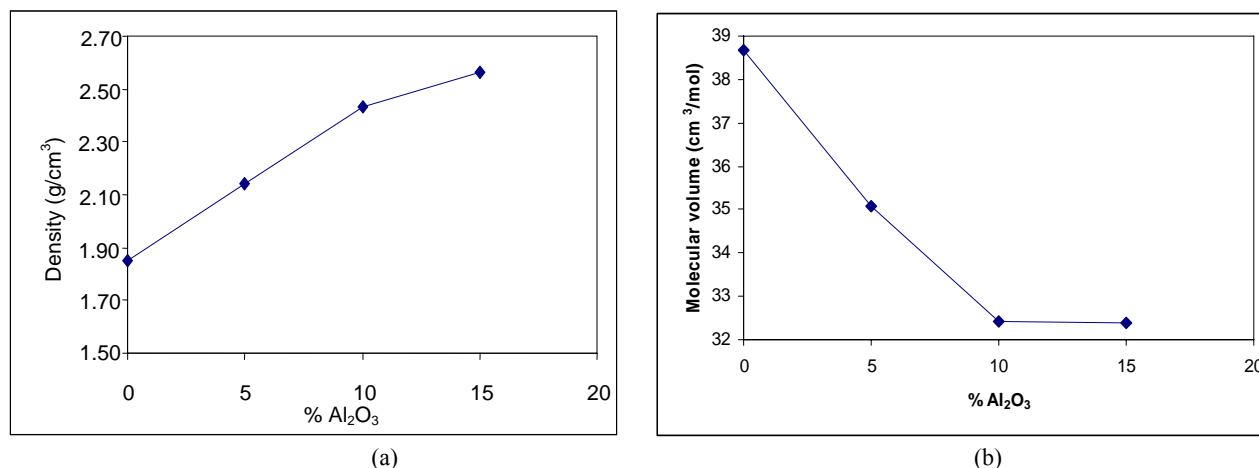


Fig. 3 Variation of  $\rho$  and  $V$  for obtained (BBA) glass system as a function of  $\text{Al}_2\text{O}_3$  mol%. (a) density (b) molecular volume.

Table 4 Chemical durability for obtained (BBA) glass system.

Sample	Acid solution (HCl)			Basic solution (NaOH)			Neutral solution (distilled water)		
	$M_0$ (g)	$M_1$ (g)	$\frac{M_0 - M_1}{M_0} (\%)$	$M_0$ (g)	$M_1$ (g)	$\frac{M_0 - M_1}{M_0} (\%)$	$M_0$ (g)	$M_1$ (g)	$\frac{M_0 - M_1}{M_0} (\%)$
G <sub>1</sub>	2.759	1.640	40.55	4.271	3.940	92.74	2.139	0.259	87.89
G <sub>2</sub>	1.443	0.775	46.29	2.228	0.380	89.94	2.382	0.622	73.88
G <sub>3</sub>	2.211	1.715	22.43	4.256	0.947	77.74	4.752	3.264	31.31
G <sub>4</sub>	1.262	1.235	2.13	2.508	1.652	34.13	2.691	1.865	30.69

**Table 5** Theoretical properties calculated.

	Properties											
	$\alpha$ in % mass	$n_d$	$d$	$E$	$\sigma$	$\rho$	$\delta t$	$\delta c$	$C_p$	$\lambda c$	$\varepsilon$	$\alpha$ in % molar
G1	0.813	1.48	0.006	252.5	94.5	1.95	64.25	857.5	0.219	0.04	5.84	0.43
G2	2.093	1.49	0.007	307.5	136.0	2.11	62.75	820	0.21	0.045	6.71	1.84
G3	3.364	1.51	0.007	362.5	177.5	2.26	61.25	782.5	0.201	0.050	6.96	3.15
G4	4.639	1.53	0.008	410	219.0	2.45	59.75	745	0.192	0.056	7.60	4.21

### 3.5 Theoretical Calcul of Properties

The results from the calculated properties are given in Table 5. According to Winkelmann and Appen, it is observed that the expansion values are respectively low in the range of 20-100 °C and 20-400 °C due to the presence of boron oxide  $\text{B}_2\text{O}_3$  which generally reduces the coefficient of thermal expansion  $\alpha$  of glasses [15]. The  $\alpha$  values increases (which may be due to the decrease in boron oxide and the presence of BaO and  $\text{Al}_2\text{O}_3$ ) and the formation of non-bridging oxygen. Results by Winkelmann and Appen are slightly different and are almost similar. The density increases with the addition of barium and alumina, as both ions have relatively large molar masses.

The presence of boron oxide also increases the refractive index of glasses which explains the high value of the refractive index calculated. Even alkaline earth oxides also contribute to the elevation of the refractive index and dispersion. With the reduction of other oxides such as boron oxide and the increase of barium and alumina the amount of non-bridging oxygen increases and this leads to an increase in the elastic modulus  $E$  and Poisson's ratio. Note that the compressive strength  $\sigma_c$  is higher than the tensile strength  $\sigma_t$  [14].

The mechanical strength of glass increases with the strength of the bonds of the glass structure. The samples prepared show low values of mechanical strength, the structure is not rigid enough, and with creation of NBO oxygen, there is a decrease in tensile strength and compression. Ordinary glasses generally are characterized by a specific heat value of about 0.8 CP (J/gk) by boric oxide increases against the property in the glasses. And with the decrease in  $\text{B}_2\text{O}_3$

and higher alumina, there was decrease in specific heat value. With the addition of  $\text{Al}_2\text{O}_3$ , there was a slight increase in thermal conductivity  $e$ , non-bridging oxygen ions increase the heat transfer inside the glass.

A decrease in the surface tension value was observed in systems of binary or ternary glasses of borates, alkaline earth oxides are important to the surface. There is a relationship between the primitive and the index of refraction as a decisive influence on the polarizability of oxygen ions. When, the  $\text{B}_2\text{O}_3$  content decrease, an increase of permittivity values with polarizability of ions (non-bridging oxygen ions) was observed with the same manner like the index of refraction.

## 4. Conclusions

In the studied glass system BaO- $\text{Al}_2\text{O}_3$ - $\text{B}_2\text{O}_3$  samples have a transparent appearance (homogeneous), except that the sample G1 is heterogeneous and presents phase separation at binary BaO- $\text{B}_2\text{O}_3$  glass, then the phase separation disappears with addition of more amount of alumina.

The addition of the alumina at the same time as some barium in the component  $\text{B}_2\text{O}_3$  contributed to the creation of non-bridging oxygen. Although the temperatures of transitions glassy  $T_g$  and softening  $T_s$  increased (because of the increase of the viscosity of the bath), Also densities and the coefficients of expansion increased. The creation of non-bridging oxygen led to an increase of the module of elasticity and this fact the reduction in the mechanical properties as the traction resistance and to the compression and the Poisson's ratio but there is an increase in chemical properties. The increase of the polarizability of the

ions oxygen has to lead to the increase of the refractive index and the dispersal as well as the electric permittivity. The properties of transfer increased as the conduction of heat and the superficial tension on the other hand the specific heat decreased.

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